

Origin of the Yellow Color of Complex Nickel Oxides

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Single-crystal optical absorption spectra of NiO, NiTiO₃, NiWO₄, NiV₂O₆, NiNb₂O₆, Ni₂SiO₄, Ni₃V₂O₈, LiNiPO₄, Li₂Ni₂Mo₃O₁₂, SrNiTeO₆, LiScSiO₄:Ni, MgSiO₃:Ni, and (Mg,Ni)₂SiO₄ are presented for the purpose of comparing the spectra of yellow and green Ni²⁺ compounds. Powder spectra of NiTiO₃, NiWO₄, NiV₂O₆, NiNb₂O₆, and Ni₃V₂O₈ in the ultraviolet region help elucidate the more intense charge transfer bands. Bright yellow color results when Ni²⁺ is in a six-coordinated site significantly distorted from octahedral symmetry. Increased absorption intensity occurs when the metal ion *d-d* bands are in proximity to an ultraviolet charge transfer band.

Introduction

Ni²⁺ in sixfold coordination in oxides generally produces green colors. After observing that Ni²⁺ is bright yellow in the distorted *M*(2) site of LiNiPO₄ of the olivine structure and in orthopyroxenes, where it apparently is in the large and distorted *M*(2) site, we initiated a study of Ni²⁺ in a variety of compounds to determine if the size of the Ni²⁺ site, its distortion, or other factors were responsible for yellow nickel compounds. Materials examined included those with Ni²⁺ in mineral structures, (three olivines and one pyroxene), and the synthetic compounds NiTiO₃, LiNiPO₄, NiWO₄, NiNb₂O₆, Sr₂NiTeO₆, Li₂Ni₂Mo₃O₁₂, Ni₃V₂O₈, and NiV₂O₆. Emphasis was

placed on single-crystal measurements which recovered both spectroscopic anisotropy and the absolute intensity of absorption and on powder reflectance measurements which allowed intensity measurements in the highly absorbing uv region between 260 and 450 nm.

Experimental Details

The olivines Ni₂SiO₄, (Mg,Ni)₂SiO₄, and LiScSiO₄:Ni, and the orthopyroxene MgSiO₃:Ni were grown by Jun Ito from a lithium molybdate-vanadate melt (1, 2). Electron microprobe analyses showed the following components by weight percent: (Mg,Ni)₂SiO₄:NiO, 46.06; FeO, 0.06; Al₂O₃, 0.12; V₂O₃, 0.49; LiScSiO₄:Ni: NiO, 7.03; V₂O₃, 0.08; FeO, 0.03; MgSiO₃:Ni: NiO, 4.18. NiO in the form of emerald green octahedra was grown from a

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concentrated KOH solution in a gold tube at 700°C and 3000 atm. Dark red crystals of NiTiO₃ were grown from a Pb-vanadate flux as described by Garton *et al.* (3). The larger crystals contain flux inclusions of an unidentified nature. An emission spectroscopic analysis showed the major impurities to be V, 0.2–1%; Fe, 0.1–0.5%; and Pb, <500 ppm. We feel the good agreement between the powder diffuse reflectance data and single-crystal absorption data eliminates the possibility that the major absorption peaks have these impurities as their origin. Yellow-orange LiNiPO₄ crystals were grown by the method of Zambonini and Malossi (4). Yellow-brown crystals of Ni₃V₂O₈ were grown from a KVO₃ flux. Brown crystals of NiWO₄ were grown from a Na₂WO₄ flux according to the method of Wanklyn (5). Yellow crystals of NiNb₂O₆ and yellow-brown crystals of Li₂Ni₂Mo₃O₁₂ were grown according to Wanklyn *et al.* (6). Yellow-brown crystals of Sr₂NiTeO₆ containing dark inclusions were grown according to Kohl *et al.* (7). Dark red crystals of NiV₂O₆ were grown using excess V₂O₅. Phase identification was by X-ray powder diffraction; cell dimensions of the various compounds refined from Guinier data are included in Table I with crystal growth details.

Oriented single-crystal spectra were measured with calcite polarizers and a Cary 17I spectrophotometer. Spectra of anisotropic crystals were obtained with the vibration direction of the incident light oriented along the principal directions of the refractive index indicatrix (corresponding to the α , β , and γ or ϵ and ω refractive indices). The relationships among optical and crystallographic directions are indicated in the figure captions. Procedures for sample preparation and thickness determination are similar to those previously reported (8). For those bands which are superimposed upon a rapidly rising background from the tail of an ultraviolet

band, a subjective estimate of the background was subtracted before arriving at ϵ values. All ϵ values are in units of liters per mole-centimeter, calculated per Ni²⁺ ion, where the concentration was calculated using densities obtained via X-ray cell parameters.

Spectra were also obtained from powder samples of NiWO₄, NiNb₂O₆, NiTiO₃, Ni₃V₂O₈, and NiV₂O₆. Powder specimens were prepared by solid state reaction. NiWO₄ when prepared from NiCl₂ · H₂O and Na₂WO₄ · 2H₂O has a dirty brown color; when prepared from Ni(OH)₂ and WO₃, NiWO₄ is bright yellow.

The absolute intrinsic absorption coefficients (α , in units of cm⁻¹) of some of the Ni compounds were determined by the dilution method of diffuse reflectance (9). Data reduction was based on the assumptions of the Kubelka–Munk theory (10). Samples were made by (1) grinding, (2) classifying to remove particles large compared with α^{-1} in spectral regions of interest, and (3) mixing the fines with a large excess of white pigmentary BaSO₄ under conditions that prevented agglomeration of colored pigment particles with each other. More details on sample preparation and data analysis have been given elsewhere (11). The method permits determination of larger absorption coefficients ($\alpha \geq 10^4$ cm⁻¹) than can readily be measured in direct transmission spectroscopy. In this work, the method is used largely for determining the strength and position of the first charge transfer band.

Results

Single-Crystal Spectra

The bulk color of several of the Ni²⁺ materials is determined primarily by the location of the transmission window defined by the wings of the absorption bands in the 800- and 400-nm regions (Figs. 1–13, Table II). For the green compounds,

TABLE I
CRYSTAL GROWTH CONDITIONS AND CELL DIMENSIONS OF Ni²⁺ COMPOUNDS

Product	Starting compositions	Initial temp. (°C)	Holding time (hr)	Final temp. (°C)	Cooling rate (°C/hr)	Appearance, size, cell dimensions
NiTiO ₃	2.8 g NiO, 3.0 g TiO ₂ , 51.5 g PbO, 22.2 g V ₂ O ₅	1320	14	950	1	black platelets 0.1 × 5 mm <i>a</i> = 5.0207(5) Å <i>c</i> = 13.785(2) Å
LiNiPO ₄	4 g LiCl, 4 g Li ₃ PO ₄ , 4 g Ni(OH) ₂ , 4 g NH ₄ H ₂ PO ₄	1000	4	300	5	yellow-orange crystals 0.3–0.5 mm <i>a</i> = 10.038(1) Å <i>b</i> = 5.858(1) Å <i>c</i> = 4.680(1) Å
NiWO ₄	3 g NiO, 10 g WO ₃ , 53.4 g Na ₂ WO ₄	1250	15	600	2	brown crystals 3–4 mm
NiNb ₂ O ₆	6.6 g NiO, 20 g Nb ₂ O ₅ , 33 g Na ₄ B ₂ O ₇	1250	8	500	2	yellow-brown crystals 2–3 mm <i>a</i> = 14.034(2) Å <i>b</i> = 5.6821(5) Å <i>c</i> = 5.0220(5) Å
Sr ₂ NiTeO ₆	6.8 g Sr ₂ NiTeO ₆ , 117 g SrCl ₂	1150	24	800	2	yellow-brown crystals 0.3–0.5 mm <i>a</i> = 9.654(3) Å <i>b</i> = 5.613(1) Å <i>c</i> = 5.591(2) Å <i>β</i> = 54.70(2)°
Li ₂ Ni ₂ Mo ₃ O ₁₂	2 g NiO 1.8 g Li ₂ CO ₃ , 10.8 g MoO ₃	1300	8	650	2	yellow-brown crystals ½ × ½ × 2 mm <i>a</i> = 10.410(2) Å <i>b</i> = 17.500(2) Å <i>c</i> = 5.075(1) Å
Ni ₃ V ₂ O ₈	1.611 g NiO, 3.923 g V ₂ O ₅ , 4.466 g KVO ₃	1000	4	500	10	tan needles 0.1 × 0.2 × 2 mm <i>a</i> = 8.239(1) Å <i>b</i> = 11.387(1) Å <i>c</i> = 5.930(1) Å
NiV ₂ O ₆	1.704 g NiO, 8.296 g V ₂ O ₅	800	4	500	10	dark red transparent platelets 0.1 × 0.2 × 0.3 mm <i>a</i> = 7.130(1) Å <i>b</i> = 8.820(2) Å <i>c</i> = 4.791(5) Å <i>α</i> = 90.15(2)° <i>β</i> = 94.19(2)° <i>γ</i> = 102.15(2)°

the transmission window is centered in the region 510–550 nm (e.g., ~524 nm in NiO and ~540 nm in both (Mg,Ni)₂SiO₄ and Ni₂SiO₄). For the bright yellow compounds it is centered in the region 570–580 nm

(e.g., ~573 nm in LiNiPO₄, 570–580 nm for MgSiO₃:Ni, and ~580 nm for LiScSiO₄:Ni). For the orange to orange-red compounds it is centered at longer wavelengths (e.g., ~590 nm in Ni₃V₂O₈).

TABLE II
 ABSORPTION BAND POSITIONS^a

		Spectral region								
		1400 nm		800 nm			400 nm			
NiO	unpol.		1142		710	643	560	464	415	378
NiTiO ₃	c		1330		833	800	745			441
	⊥c		1346		840	798	742			499
NiWO ₄	α	1530	1385		819	738			496	438
	β		1341		837	739	517			443
	γ		1317		836	739				452
NiV ₂ O ₆	unpol.		1285		883	767				
NiNb ₂ O ₆	α		1363		826	733	508			444
	β		1428		815	729	506	485		440
	γ		1333		823	733	509			449
Ni ₂ SiO ₄	α		1255		788	707	500	474		420
	β	1482	1372		774	706	621	474	415	376
	γ		1247	1100	786	708		472	423	378
Ni ₃ V ₂ O ₈	α		1270			733				
	β		1277			741				
	γ	1540	1260		843	732		534		
LiNiPO ₄	α		1620		806	704			473	424
	β	1775	1450	1304	811	704	504	472		426
	γ		1297		798	702	512	466		424
Li ₂ Ni ₂ Mo ₃ O ₁₂	α		1389		810	732				434
	β		1303		813	735				440
	γ		1352		811	738	526			434
Sr ₂ NiTeO ₆	unpol.	1621	1432		865	732	702	516		461
LiScSiO ₄ :Ni	α		1367		833	712	522	485		427
	β		1705	1378	840	714	676			425
	γ		1330		840	713		487		431
MgSiO ₃ :Ni	α	2025	1389		804				462	409
	β	2050	1483			771	670		464	408
	γ	2050	1540	937		771			464	408
(Mg, Ni)SiO ₄	α		1272		796	708	630	505	479	422
	β	1435	1299	1104	777	709	644			419
	γ		1288	1208	793	709	632		478	425

^a In cases where the bands are broad, overlapping, or asymmetric, the tabulated value represents our best estimate of the band maximum. Several very weak, presumably spin-forbidden, bands which are not tabulated above can be observed in the data. All bands are tabulated in units of nm.

The exact wavelength of the transmission maximum will depend on the polarization for the low-symmetry compounds.

Other compounds, such as NiTiO₃,

Sr₂NiTeO₆, and NiV₂O₆, either do not have a transmission window centered in the visible portion of the spectrum, or it is centered so far into the red that the wings of the long-

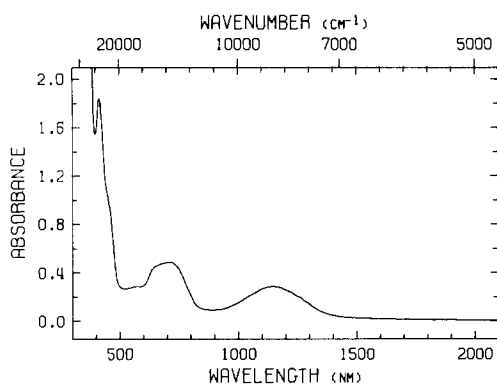


FIG. 1. Absorption spectrum of NiO plotted as 13.3 μm thick.

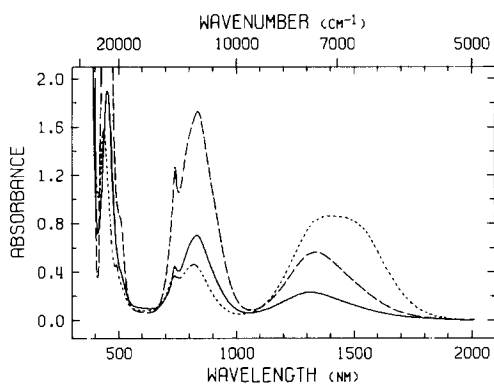


FIG. 3. Absorption spectrum of NiWO₄ plotted as 24.0 μm thick. ---, $\alpha \approx c$; —, $\beta = b$; — · —, $\gamma \approx a$; $\alpha \wedge c = 13^\circ$.

wavelength band do not appreciably influence the color. Likewise, the long-wavelength wings of NiWO₄, NiNb₂O₆, and Li₂Ni₂Mo₃O₁₂ are too weak at visible wavelengths to have much influence on color. In these cases, the color is caused predominantly by the tail of the absorption bands centered at less than 400 nm. The net result is a yellow-orange to red color for thin samples. The bulk color of millimeter-sized crystals of several of the compounds with this type of spectrum (e.g., NiTiO₃ and NiV₂O₆) is black because of the overall high level of absorption throughout the visible region.

The color of several of the compounds is dependent upon the presence of medium-intensity, low-energy components of the 400-nm region absorption system. These occur at ~ 464 nm in the MgSiO₃:Ni spectra, ~ 472 and ~ 512 in the LiNiPO₄ spectra, and ~ 508 nm in the NiNb₂O₆ spectra. These components also occur in the spectra of NiWO₄, Sr₂NiTeO₆, and LiScSiO₄:Ni. In the spectra of several of the compounds, multiple components are found on the long-wavelength side of the primary band near 400 nm. Our estimates of the ϵ values of these bands range from 2.0 for Li₂Ni₂Mo₃O₁₂ to 14.1 and 17.3 for LiNiPO₄

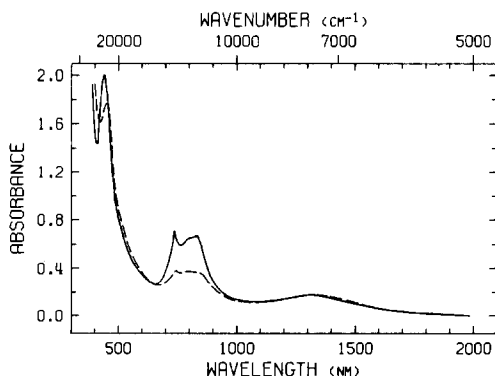


FIG. 2. Absorption spectrum of NiTiO₃ plotted as 5.0 μm thick. Solid line, polarized $\parallel c$; dashed line, polarized $\perp c$.

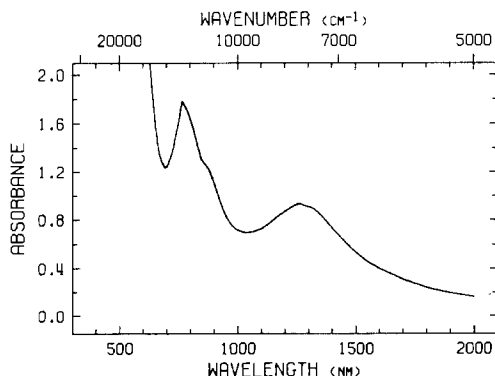


FIG. 4. Absorption spectrum of NiV₂O₆ plotted as 33 μm thick. Unpolarized; orientation of crystal not determined.

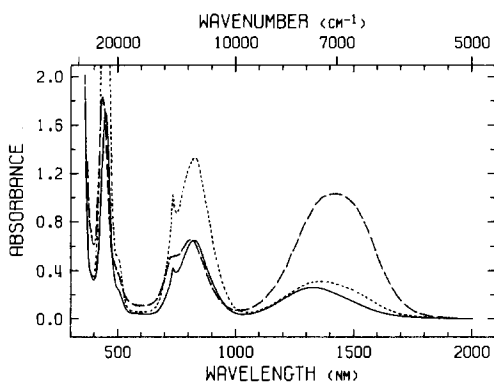


FIG. 5. Absorption spectrum of NiNb₂O₆ plotted as 60 μm thick. ---, α = b; —, β = a; — · —, γ = c.

and MgSiO₃:Ni, respectively. These last two materials have the brightest yellow color of the substances studied.

Diffuse Reflectance Spectra

Intrinsic absorption spectra obtained from diffuse reflectance for the materials tested are given in Fig. 14. Strong, charge transfer transitions dominate the colors of Ni₃V₂O₈ and NiV₂O₈ in the visible. The three compounds with their charge transfer edge in the near uv are listed in Table 4 together with the wavelength at which $\alpha = 10^5 \text{ cm}^{-1}$. Also shown are suitable averages of the single-crystal ϵ values at $\sim 400 \text{ nm}$. For the uniaxial case, the average is $(2\epsilon_{\perp} + \epsilon_{\parallel})/3$. For biaxial systems the average of the

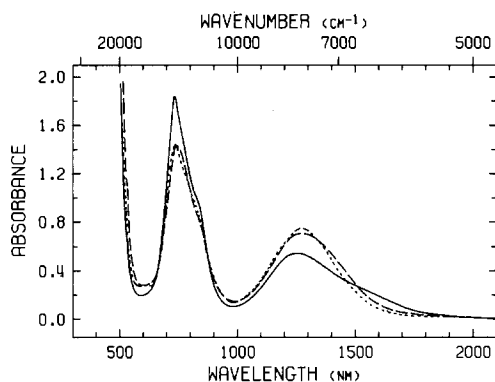


FIG. 7. Absorption spectrum of Ni₃V₂O₈ plotted as 40 μm thick. ---, α = b; —, β = c; — · —, γ = a.

three ϵ values is given. We see that for the three compounds examined, the larger ϵ values are associated with closer proximity to the uv charge transfer transitions.

Accuracy in the dilution method of diffuse reflectance requires that the concentration of colored material in the white diluent be not much larger than 1% and that the reflectance of the mixture lie in the region $0.2 < R < 0.6$. While it is well suited to measurements of absorption coefficients in the range $\alpha > 10^4 \text{ cm}^{-1}$, the method is not as good for measurements of $\alpha < 10^3 \text{ cm}^{-1}$, particularly in mixtures designed for measuring the stronger charge transfer transitions. Accordingly, the *d-d* transitions that form the main subject of the paper are only

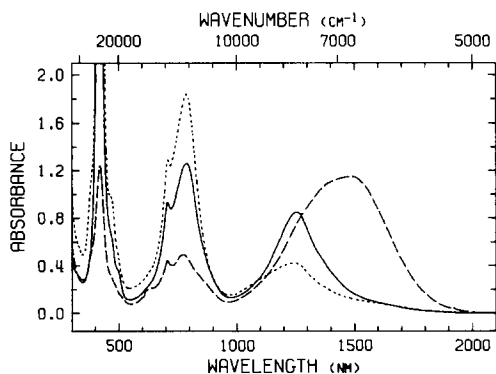


FIG. 6. Absorption spectrum of Ni₂SiO₄ plotted as 35 μm thick. ---, α = b; —, β = c; — · —, γ = a.

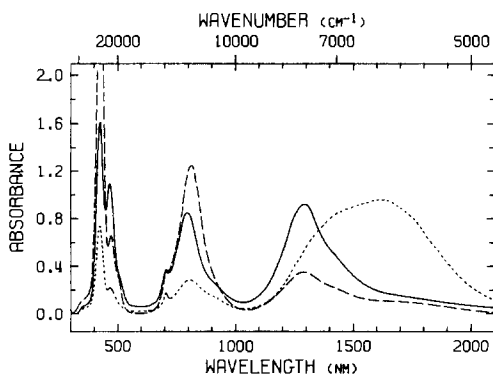


FIG. 8. Absorption spectrum of LiNiPO₄ plotted as 30 μm thick. ---, α = c; —, β = a; — · —, γ = b.

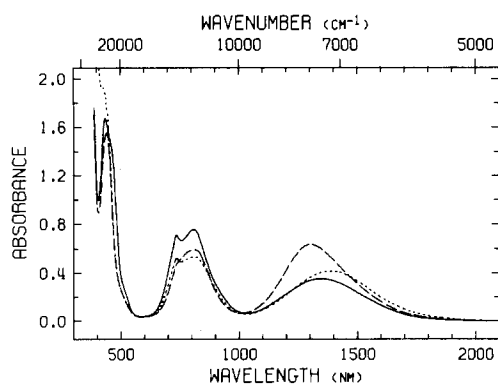


FIG. 9. Absorption spectrum of $\text{Li}_2\text{Ni}_2\text{Mo}_3\text{O}_{12}$ plotted as 50 μm thick. ---, $\alpha = c$; —, $\beta = a$; — · —, $\gamma = b$.

weakly observable in the diffuse reflectance spectra taken to determine the position of the first charge transfer band. Nevertheless, the relative increase above background observed in the spectra of Fig. 14 near 450 nm is consistent with the relative single-crystal minimum ϵ values there.

Electronic Origin of the Absorption Bands

The general features of Ni^{2+} absorption have been described in terms of ligand field theory. In a rigorously octahedral environment, Ni^{2+} will have three prominent absorption bands, arising from transitions from the ${}^3A_{2g}$ ground state to the ${}^3T_{2g}$,

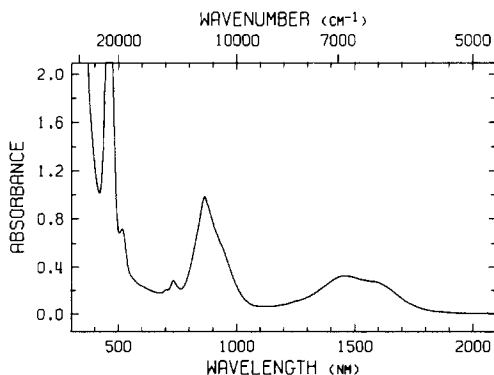


FIG. 10. Absorption spectrum of $\text{Sr}_2\text{NiTeO}_6$ plotted as 100 μm thick. Unpolarized.

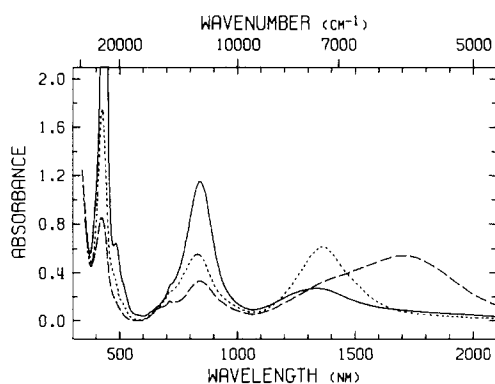


FIG. 11. Absorption spectrum of $\text{LiScSiO}_4:\text{Ni}$ plotted as 360 μm thick. ---, $\alpha = b$; —, $\beta = c$; — · —, $\gamma = a$.

and ${}^3T_{1g}(F)$, and ${}^3T_{1g}(G)$ states. These correspond to the prominent absorption bands in the 1400, 800, and 400-nm regions, respectively (12, 13). Reinen (14, 15) examined the reflection spectra of a variety of compounds with Ni^{2+} in sixfold coordination and concluded that transitions to the singlet states account for the remaining minor features in the spectra. Strong spin-orbit coupling together with antiferromagnetic interactions is believed to increase the intensity of the transition to the 1E_g state at ~ 720 –740 nm such that it becomes more intense than the spin-allowed transition to the ${}^3T_{1g}$ state in the case of NiO. Likewise the transition to the ${}^1T_{1g}$ state at 378 nm is more intense than the 415 nm ${}^3T_{1g}$ band in NiO.

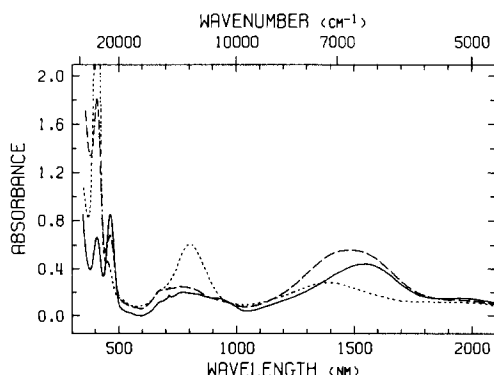


FIG. 12. Absorption spectrum of $\text{MgSiO}_3:\text{Ni}$ plotted as 250 μm thick. ---, $\alpha = b$; —, $\beta = a$; — · —, $\gamma = c$.

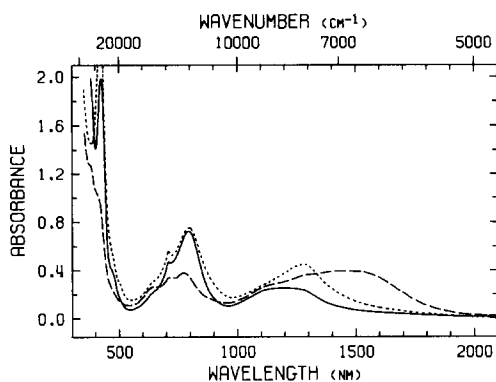


FIG. 13. Absorption spectrum of $(\text{Mg}, \text{Ni})_2\text{SiO}_4$ plotted as $42 \mu\text{m}$ thick. ---, $\alpha = a$; —, $\beta = c$; — · —, $\gamma = b$.

Intensities

The greatest absorption band intensities in the 400-nm region are associated with metal ion absorption bands in proximity to an ultraviolet charge transfer tail (e.g., NiTiO_3) or with Ni^{2+} in a distorted site (e.g., $\text{MgSiO}_3:\text{Ni}$). The same effects operate in the 800-nm region, but are not clearly identifiable in the 1400-nm region. The most intense bands, those of NiTiO_3 , are about an order of magnitude more intense than those of reference Ni^{2+} compounds in aqueous solutions and NiO . Proximity to an ultraviolet charge transfer tail has been recognized to be an important factor in determining the intensities of electronic transitions of metal ions in general through the mechanism of "intensity stealing" (16).

The extent to which the ultraviolet charge transfer band extends into the visible region depends in part on the ion with which the Ni^{2+} shares oxygen ions. Easily reduced ions such as V^{5+} and Ti^{4+} produce a charge transfer tail which extends well into the visible region, whereas difficultly reduced ions such as P^{5+} and Si^{4+} do not. Intermediate tails are observed for W^{6+} , Mo^{6+} , and Nb^{5+} . The point at which the absorption on the charge transfer tail rises to $200/\text{cm}$ is included in Table III. The

contribution of overlapping Ni^{2+} bands was visually subtracted to obtain these values. [The point at which it rises to $10^5/\text{cm}$ is shown in Table IV.] For systems with charge transfer bands extending over much of the visible region, as $\text{Ni}_3\text{V}_2\text{O}_8$ and NiV_2O_6 (Fig. 14), the color of the material is dominated by the influence of the tail.

Site Symmetry

Site symmetry influences the color of Ni^{2+} compounds. The degeneracy of the ${}^3T_{1g}$, ${}^3T_{2g}$, and 3E_g states of Ni^{2+} in octahedral geometry are lifted and multiple-absorption bands occur when Ni^{2+} is situated in a coordination environment of low symmetry. Bands split by low symmetry will show peak wavelengths which are polarization dependent (17). Such splittings are present in the spectra of most of the compounds we studied, and are especially pronounced in the spectra of compounds with the olivine and orthopyroxene structure. The compounds with the brightest yellow color, LiNiPO_4 and $\text{MgSiO}_3:\text{Ni}$, also have the most pronounced splitting of the absorption bands.

Band intensities often increase in response to low site symmetry. The high

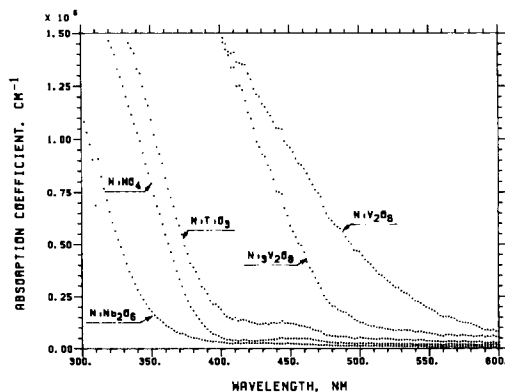


FIG. 14. Absorption spectra of NiNb_2O_6 , NiWO_4 , NiTiO_3 , $\text{Ni}_3\text{V}_2\text{O}_8$, and NiVO_6 powders obtained from diffuse reflectance.

TABLE III
 ϵ VALUES OF Ni^{2+} COMPOUNDS (liters/mole · cm)

Compound	Polarization	Spectral region ^a			C.T. abs = 200/cm ^b (nm)
		1400 nm	800 nm	400 nm	
NiO	unpol.	2.1	3.7	12.6	390
NiTiO ₃	$\perp c$	7.5	13	88	600
	$\parallel c$	7.9	32	102	590
NiWO ₄	α	3.0	9.8	28.4	422
	β	7.9	25.0	46.7	417
	γ	16.5	6.2	22.0	415
NiV ₂ O ₆	unpol.	9.5	20	n.d.	650
NiNb ₂ O ₆	α	3.0	12.8	30.9	365
	β	10.0	5.9	17.6	373
	γ	2.5	6.1	16.7	367
Ni ₂ SiO ₄	α	5.0	6.8	21.1	280
	β	6.9	2.5	7.5	280
	γ	2.3	9.9	28.6	280
Ni ₃ V ₂ O ₈	α	7.3	13.8	n.d.	520
	β	6.6	14.0	n.d.	520
	γ	4.9	17.7	n.d.	520
LiNiPO ₄	α	13.5	3.7	9.9	<300
	β	4.8	17.6	37.1	<300
	γ	12.5	11.5	22.0	<300
Li ₂ Ni ₂ Mo ₃ O ₁₂	α	6.4	7.8	—	~420
	β	8.6	7.8	21.4	~400
	γ	4.5	9.6	21.5	~400
Sr ₂ NiTeO ₆	unpol.	2.4	7.3	16.9	373
LiScSiO ₄ : Ni	α	5.4	5.0	15.6	310
	β	4.9	3.0	7.7	310
	γ	2.0	9.8	32	310
MgSiO ₃ : Ni	α	5.9	12.5	67	320
	β	11.6	4.7	34	320
	γ	9.0	4.2	10	320
(Mg, Ni) ₂ SiO ₄	α	2.8	4.3	14.9	340
	β	3.0	2.6	2.4	390
	γ	1.7	5.0	18.1	410

Note. n.d. = not determined.

^a The ϵ values under these headings refer to the intensity of the most intense band in the region near 1400, 800, and 400 nm. They are computed disregarding multiple-site occupancy of nickel.

^b The wavelength at which the ultraviolet charge transfer tail reaches an absorbance of 200/cm of crystal thickness.

intensities of the LiNiPO₄ absorptions are an example. The high intensity of the 457-nm spin-forbidden band of Ni²⁺ clinopyrox-

ene previously noted by White *et al.* (18) is analogous to the intensity of the 464-nm band in MgSiO₃ : Ni.

TABLE IV
 ϵ VALUES AT 400 nm COMPARED WITH POSITION OF
 CHARGE TRANSFER BAND

	ϵ at 400 nm (l mole ⁻¹ cm ⁻¹)	$\alpha = 10^5$ cm ⁻¹ (nm)
NiTiO ₃	92.7	352
NiWO ₄	32.4	340
NiNb ₂ O ₆	21.7	305

Correlations

Correlations were examined between structural, ligand field, and other spectroscopic parameters. In general, no strong correlations were established. For Ni²⁺ in a rigorously octahedral site, the value of the ligand field parameter Dq is equal to 1/10th the energy of the transition to the lowest energy ³ T_{2g} state. In a site of low symmetry, the transition to ³ T_{2g} is split into up to three components. We defined $10Dq$ to be equal to the average of the transition energies of the three components in the 1400-nm region.¹ In some cases our values differ from those obtained from diffuse reflectance spectra because we have included weak, low-energy components which were not identified in previous studies. Faye had presented a linear correlation between the average Ni–O distance and Dq for a series of compounds containing six-coordinated Ni²⁺ (19). Unfortunately, this correlation fails for NiWO₄, NiTiO₃, Ni₃V₂O₈, NiNb₂O₆, Sr₂NiTeO₆, and Li₂Ni₂Mo₃O₁₂. Likewise, Dq does not correlate with a parameter which measures the distortion of the Ni²⁺ site defined by:

$$\text{Distortion Parameter} = \frac{1}{3} \sum \{(R_i - \bar{R})/\bar{R}\}^2,$$

where \bar{R} is the mean Ni–O distance and R_i 's are the individual Ni–O distances. Furthermore, Dq does not correlate well

¹ For biaxial crystals $10Dq = \frac{1}{3} [E(\alpha) + E(\beta) + E(\gamma)]$. For uniaxial NiTiO₃, $10Dq = \frac{1}{3} [2E(\perp c) + E(\parallel c)]$.

with the ϵ values. Determination of ligand field parameters is complicated by multiple-site occupancy in some compounds. In Ni₂SiO₄, for example, Ni²⁺ occurs in two sites, $M1$ and $M2$, which are crystallographically distinct. (20). The overlapping contributions from Ni²⁺ in both sites are present in the spectrum. The overlap of bands from multiple sites and low site symmetry both render detailed calculation of ligand field parameters impractical.

Comparison to Other Studies

The details of intensities and band shapes of the NiO single-crystal spectra are different from that of MgO:Ni²⁺ crystal data (12, 21), particularly with regard to the presence of the 560-nm band, which was also observed in NiO by Newman and Chrenko (22). Most previous studies of nickel compounds have been by reflection spectroscopy. We frequently find differences in both band positions and relative intensities between the single-crystal data and those calculated from reflectance data on concentrated powers. These studies include: NiTiO₃ (14), Ni₂SiO₄ (23, 24), and (Mg,Ni)₂SiO₄ (23). Overall good agreement was observed between our Sr₂NiTeO₆ single-crystal data and the reflectance data of Kohl (25). Spectroscopic crystal field parameters only have been reported for NiNb₂O₆ (15). White *et al.* (18) concluded that Ni²⁺ was situated on the larger, distorted $M2$ site of the yellow, monoclinic pyroxene MgSiO₃:Ni. Their conclusion was based on the small value of Dq and the large value of the Racah parameter B . Similar values of these parameters are calculated from our MgSiO₃:Ni spectrum. The low energy components near 2000 nm indicate greater than usual splitting of the lowest ³ T_{1g} state and are further evidences for the occupancy by Ni²⁺ of the distorted $M2$ site.

Reinen (23) concluded from reflectance

spectra that although Ni^{2+} preferentially orders on $M1$ in $(\text{Mg,Ni})_2\text{SiO}_4$, low-energy components on the three primary band systems were from Ni^{2+} in the $M2$ site. In the spectrum of Ni_2SiO_4 , Wood (24) identified contributions from both the $M1$ and $M2$ sites; however, our single-crystal spectra differ significantly from his fit of a reflectance spectrum.

Conclusions

Bright yellow Ni^{2+} compounds can occur as a consequence of nickel entering six-coordinated sites significantly distorted from octahedral geometry. The color is a result of electronic transitions localized on the Ni^{2+} . Yellow compounds also result from intense charge transfer tails centered in the ultraviolet. These are associated with easily reduced cations. Higher absorption coefficients are associated with proximity to uv charge transfer bands through the mechanism of "intensity stealing."

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